

# Structural and Magnetic Properties of Zn<sup>2+</sup> Doped Cobalt Ferrite Nanoparticles Synthesized by Sol-gel Auto Combustion Method

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**Abstract:** Zinc (Zn<sup>2+</sup>) doped spinel cobalt ferrites nanoparticles having composition (Zn<sub>x</sub>Co<sub>1-x</sub>Fe<sub>2</sub>O<sub>4</sub> where x = 0.2, 0.4) were synthesized by sol-gel auto combustion method. The structural and magnetic properties were characterized via X-ray diffraction (XRD) technique and Pulse field hysteresis loop tracer. The average crystallite size (D) was found to decrease from 32.15 nm to 24.31 nm with increasing the Zn<sup>2+</sup>. Magnetization field loops reveal ferromagnetic behavior, lower hysteresis loop observed for higher Zn content. The magnetic properties are remarkably influenced with Zn doping; saturation magnetization (Ms) decreases then increases while both coercivity (HC) and remanent magnetization (Mr) decrease with increase in Zn<sup>2+</sup> content, which was associated with preferential site occupancy and the change in average crystallite size.

**Keywords:** XRD, Magnetization, Spinel ferrite, nanoparticles

## 1. Introduction

Spinel ferrite nanoparticles have general formula of MFe<sub>2</sub>O<sub>4</sub>, where M is a metal cation with charge such as Mn, Fe, Cu, Ni and Zn. These nanoparticles have unique magnetic and electrical properties and thus have been utilized in many applications such as, magnetic recording and sensing, information storage, catalysis, gas sensors, battery applications [1-3]. This is because of various interesting properties possessed by the materials, e.g., strong anisotropy, high saturation, coercivity, etc. [4]. The magnetic properties of the CoFe<sub>2</sub>O<sub>4</sub> are greatly affected by the size of the particles, by cation substitution and distribution [5]. ZnFe<sub>2</sub>O<sub>4</sub> is a normal spinel with all the Fe<sup>3+</sup> ions in the sites and all the Zn<sup>2+</sup> ions in the (A) sites, whereas CoFe<sub>2</sub>O<sub>4</sub> has an inverse spinel structure with the Co<sup>2+</sup> ions mainly in the sites and Fe<sup>3+</sup> ions distributed almost equally between the (A) and the sites [6]. However, ZnFe<sub>2</sub>O<sub>4</sub> with an extreme case has an inverse spinel structure. Zn<sup>2+</sup> substitution in CoFe<sub>2</sub>O<sub>4</sub> may have some distorted spinel structures depending upon the concentration of cations. Therefore, CoZnFe<sub>2</sub>O<sub>4</sub> has a mixed spinel type of structure. It is known that zinc ions (Zn<sup>2+</sup>) with diamagnetic nature are known for achieving good control over magnetic parameters in developing technologically important materials [7-9]. Substitution of magnetic (Co<sup>2+</sup>) by a nonmagnetic (Zn<sup>2+</sup>) cation in spinel ferrite phase may induce important changes in their structural, optical, magnetic, and others properties, due to the distribution of cations in between the available A and B sites. A lot of work has been reported in the literature about different types of nonmagnetic and magnetic ions doped Co-ferrites synthesized by various methods. [10-15]. In this work, we investigate the magnetic and structural properties of Zn<sup>2+</sup> ions substituted cobalt ferrite, Zn<sub>x</sub>Co<sub>1-x</sub>Fe<sub>2</sub>O<sub>4</sub> (x = 0.0, 0.2 and 0.4) synthesized by the sol gel auto-combustion method. The substitution of non-magnetic Zn<sup>2+</sup> ion is expected to modify structural and magnetic properties by altering the cation distribution between the two crystallographic (A) and [B] sites.

## 2. Experimental

The nanocrystalline zinc ions substituted cobalt ferrite nanoparticles Co<sub>1-x</sub>Zn<sub>x</sub>Fe<sub>2</sub>O<sub>4</sub> (x = 0.0–0.4) were synthesized by the sol auto combustion method. For the preparation of Co<sub>1-x</sub>Zn<sub>x</sub>Fe<sub>2</sub>O<sub>4</sub> (x = 0.0, 0.2, 0.4) nanoparticles through sol gel auto combustion method. cobalt, zinc, and iron nitrates were taken in stoichiometric proportions and then dissolved separately in de-ionized water at constant heating at 60–80 °C under continuous stirring. The glycine was used as chelating agent for combustion. The glycine solution was added separately to cobalt nitrate, zinc nitrate and iron nitrate solutions, with continuously heating and stirring; to obtain a transparent brown color glossy gel. The molar ratio of nitrates to glycine was taken according to propellant chemistry. Heating was continued at above 100 °C, the gel was then swelled and ignited with evolution of large amount of gaseous product, resulting the desired ferrite in the form of foamy powder. The obtained powders were then finally calcined at 700 °C for 5 h.

## 3. Characterization

X-ray diffraction (XRD) measurements were recorded with Ultima IV, Rigaku Corporation, Japan. by using Cu-K $\alpha$  radiation ( $\lambda=1.5418\text{\AA}$ ). The XRD patterns were recorded at room temperature in the 2 $\theta$  range of 200-800 with a step size of 0.02 and a speed of 0.5/ min. M – H of all samples was studied with help of pulse field hysteresis loop tracer (Magneta Ltd.) technique at room temperature.

## 4. Results and Discussions

### 4.1 X-ray diffraction analysis

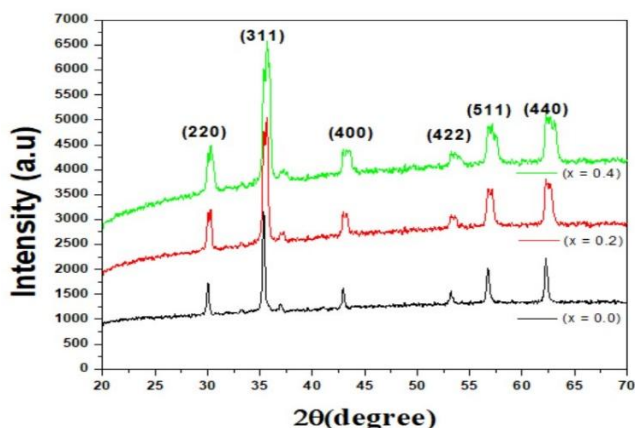
The X-ray diffraction patterns of the Co<sub>1-x</sub>Zn<sub>x</sub>Fe<sub>2</sub>O<sub>4</sub> spinel ferrite nanoparticles with x = 0.0, 0.2, 0.4 are shown in Fig. 1.0. From the X-Ray diffraction analysis data, it is found that all diffraction peaks very well match with reported in

literature [16] and (ICDD: 01-072-6799). The peak broadness can be attributed to the nanocrystalline nature of the prepared samples. The intensity of the XRD peaks is found to decrease while the peak broadening increases with increasing  $Zn^{2+}$  doping concentration. The lattice constant, crystallite size and X-ray density was calculated from the obtained XRD data and all these parameters calculated using the relation [17, 18] and is tabulated in Table 1.

**Table 1:** Lattice constant, crystallite size and X-ray density of  $Co_{1-x}Zn_xFe_2O_4$  nanoparticles

conc 'x'	Lattice parameter 'a' (Å)	Crystallite size (nm)	X-ray density (gm/cm <sup>3</sup> )
0.0	8.372	32.15	5.610
0.2	8.378	28.02	5.774
0.4	8.487	24.31	5.875

The lattice parameter is found to increase from 8.372 Å to 8.487 Å with increasing  $Zn^{2+}$  doping. The increase of lattice parameter with  $Zn^{2+}$  doping concentration can be explained on the basis of the difference in ionic radii of  $Zn^{2+}$  and  $Co^{2+}$ . The smaller ionic radius of  $Co^{2+}$  (0.58 Å) was replaced by the larger ionic radius of  $Zn^{2+}$  (0.6 Å) so the lattice parameter increased due to the expansion of the unit cell [19]. All the peaks in the diffraction pattern have been indexed and Cohen's method was used for the refinement of the lattice parameter [20]. The crystallite was calculated by using Scherrer's equation. The crystallite size was found in the range 24.32 nm to 32.15 nm. Crystallite size decreases with the increase in  $Zn^{2+}$  substitution.



**Figure 1:** The X-ray diffraction patterns of the  $Co_{1-x}Zn_xFe_2O_4$  as function of  $Zn^{2+}$  content

The reduction in particles size is may be due to the mechanically induced contraction in Co-Zn ferrite due to the deformation of  $Fe^{2+}-O^{2-}-Fe^{3+}$ . The X-ray density increases with increase in  $Zn^{2+}$  ions. This may be due to the larger atomic weight of the  $Zn^{2+}$  ions as compared to that of  $Co^{2+}$  ions.

#### 4.2. Magnetic Properties

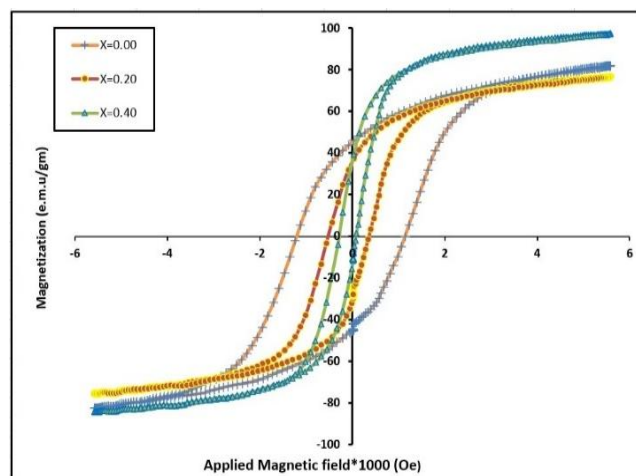
Magnetic hysteresis (M-H) loops of the prepared nanoparticles recorded at 300 K are shown in Fig. 2. From hysteresis loop, it is observed that, the  $CoFe_2O_4$  shows a ferromagnetic behavior with a large hysteresis loop. Doping with Zn ions reveals also a ferromagnetic behavior while induces important modifications in the magnetic properties; the hysteresis loop decreases drastically with Zn content.

Magnetic parameters such as saturation magnetization ( $M_s$ ), remanence magnetization ( $M_r$ ) and coercivity ( $H_c$ ), magnetic moment ( $\eta B$ ), were calculated for all samples at room temperatures and are listed in Table 2.

**Table 2:** Saturation magnetization ( $M_s$ ), remanence magnetization ( $M_r$ ), coercivity ( $H_c$ ), remanent ratio (R), Calculated ( $nB$ ) and Magnetic moment  $Co_{1-x}Zn_xFe_2O_4$  nanoparticles

Conc 'x'	$M_s$	$M_r$	$H_c$	$M_r/M_s$	$nB$ ( $\mu B$ )
	(emu/g)	(emu/g)	(Oe)		
0.0	84.15	74.56	1122	0.907	3.45
0.2	82.66	36.00	496	0.425	3.58
0.4	98.12	23.00	186	0.234	4.16

It is observed from M-H loop and tabulated data of table 2, saturation magnetization increases with increase in Zn content. This could be due to  $Zn^{2+}$  (with zero magnetic moment) replace ion on the tetrahedral (A) sites, causing the decrease of magnetic moment in the sub-lattice MA, resulting in the increase of total magnetic. It is also observed that the remanence magnetization decreases with increase in zinc content x. The decrease in remanence magnetization values can be attributed to the substitution of nonmagnetic zinc ions in cobalt ferrite [21]. M-H plot shows that the all samples show high values of coercivity confirming the nanocrystalline nature. Further, it can be observed that the coercivity decreases with increase in substitution of zinc content x.



**Figure 2:** M-H curves of the  $Co_{1-x}Zn_xFe_2O_4$  nanoparticles recorded at 300 K.

The decrease in coercivity values can be attributed to the substitution of nonmagnetic zinc ions in cobalt ferrite. Table 2 shows that the values of all magnetic parameters decrease as a function of substitution of zinc content x. It may be due to the fact that nonmagnetic zinc replaces magnetic  $Co^{2+}$  ions. The A-B magnetic interaction decreases due to substitution of zinc, as a result of which the magnetic properties decrease as a function of zinc content x [22].

#### 5. Conclusion

Zn-doped cobalt-ferrite nanoparticles, with nominal composition  $Co_{1-x}Zn_xFe_2O_4$  ( $x = 0.0, 0.2$ , and  $0.4$ ), were successfully synthesized by sol gel auto-combustion

technique. The analysis of XRD patterns reveals the formation of single-phase cubic spinel structure. The saturation magnetization increases with increase in Zn content while remanence magnetization and coercivity decrease as a function of substitution of zinc content  $x$ . It may be due to the fact that nonmagnetic zinc replaces magnetic  $\text{Co}^{2+}$  ions.

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